Two Players Make a Formidable Combination: In Situ Generated Poly(acrylic anhydride-2-methyl-acrylic acid-2-oxirane-ethyl ester-methyl methacrylate) Cross-Linking Gel Polymer Electrolyte toward 5 V High-Voltage Batteries

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Supporting Information

ABSTRACT: Electrochemical performance of high-voltage lithium batteries with high energy density is limited because of the electrolyte instability and the electrode/electrolyte interfacial reactivity. Hence, a cross-linking polymer network of poly(acrylic anhydride-2-methyl-acrylic acid-2-oxirane-ethyl ester-methyl methacrylate) (PAMM)-based electrolyte was introduced via in situ polymerization inspired by “liangian hebi”, which is a statement in a traditional Chinese Kungfu story similar to the synergetic effect of 1 + 1 > 2. A poly(acrylic anhydride) and poly(methyl methacrylate)-based system is very promising as electrolyte materials for lithium-ion batteries, in which the anhydride and acrylate groups can provide high voltage resistance and fast ionic conductivity, respectively. As a result, the cross-linking PAMM-based electrolyte possesses a significant comprehensive enhancement, including electrochemical stability window exceeding 5 V vs Li+/Li, an ionic conductivity of 6.79 × 10⁻¹⁰ S cm⁻¹ at room temperature, high mechanical strength (27.5 MPa), good flame resistance, and excellent interface compatibility with Li metal. It is also demonstrated that this gel polymer electrolyte suppresses the negative effect resulting from dissolution of Mn²⁺ ions at 25 and 55 °C. Thus, the LiNi₀.₅Mn₁.₅O₄/Li and LiNi₀.₅Mn₁.₅O₄/Li₄Ti₅O₁₂ cells using the optimized in situ polymerized cross-linking PAMM-based gel polymer electrolyte deliver stable charging/discharging profiles and excellent rate performance at room temperature and even at 55 °C. These findings suggest that the cross-linking PAMM is an intriguing candidate for 5 V class high-voltage gel polymer electrolyte toward high-energy lithium-ion batteries.

KEYWORDS: poly(acrylic anhydride), poly(methyl methacrylate), in situ polymerization, gel polymer electrolyte, high voltage lithium batteries

INTRODUCTION

Recently, high-energy-density lithium-ion batteries (LIBs) have been extensively researched to mitigate the energy issues.¹–³ Spinel LiNi₀.₅Mn₁.₅O₄, having high working voltage plateau (~4.7 V vs Li⁺/Li) and theoretical capacity (148 mAh g⁻¹), is a promising cathode material for high-energy-density LIBs.⁴–⁶ However, the conventional carbonate-based liquid electrolytes fail to meet the requirement of LiNi₀.₅Mn₁.₅O₄ practical application because of the low breakdown voltage below 4.5 V vs Li⁺/Li.⁷–⁸ Moreover, the interface reaction of Mn²⁺ ions dissolution from LiNi₀.₅Mn₁.₅O₄ cathode deteriorates the battery cycling stability seriously.¹⁹–₂⁰ As a result, various efforts have been made to achieve excellent electrolyte system in the past few years.¹⁸–ⁱ⁹ Our group has presented a series of cyano-based polymer electrolytes (Table 1) that display high voltage resistance (>4.7 V), high ionic conductivity (>1 × 10⁻³ S cm⁻¹) at room temperature, and excellent safety.¹⁶,¹⁷ In addition, the strong chelation of cyano group to transition metal ions can effectively suppress the transfer of Mn²⁺ ions from LiNi₀.₅Mn₁.₅O₄ to the opposite anode and then enhance the interface stability. Therefore, exploring novel versatile polymer electrolyte is significant to the application of LiNi₀.₅Mn₁.₅O₄ in high-energy-density LIBs.¹₄,¹₅,¹₇,¹₉ Anhydrides such as 1-proplyphosphonic acid cyclic anhydride²⁰ and glutaric anhydride²¹ are promising electrolyte additives for a high-voltage LiNi₀.₅Mn₁.₅O₄ battery system.²²,²³

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The anhydrides can oxidize prior to the carbonate-based liquid electrolytes during charging to high voltage and polymerize as a stable passivation film between cathode and electrolyte to suppress the electrolyte decomposition, transition metal dissolution, and cathode erosion. Therefore, an enhanced cycling stability and decreased self-discharge can be obtained for LiNi0.5Mn1.5O4-based LIBs. Nevertheless, its intrinsic poor conductivity has limited the further application of anhydrides. Furthermore, the addition of anhydride monomer would cause capacity loss as the anhydride might react with Li+, cross through the separator and deposit on the surface of Li foil during cycling. Poly(methyl methacrylate) (PMMA), also known as acrylic or acrylic glass, is a transparent thermoplastic resin often used in sheet form as a lightweight or alternative to glass. It can be utilized as casting resin, inks, coating component and so on. PMMA was first discovered as polymer electrolyte for LIBs application in 1985.24 The as-obtained results have shown that the ester group can make a contribution to fast electrolyte uptake.25 In addition, the segments in PMMA show superior binding with both electrodes will be achieved and enhanced mechanical strength endowed by anhydride and acrylic, respectively, but also the in situ polymerization consumed existed in ex situ polymerization methods could be avoided during in situ polymerization.28,29

In this work, cross-linking poly(acrylic anhydride-2-methyl-acrylic acid-2-oxirane-ethyl ester-methyl methacrylate) (PAMM) based electrolyte was synthesized for high voltage LIBs. To evaluate the advantageous performance of in situ cross-linking PAMM, the electrolyte of PMMA was presented for a fair comparison. We performed a comprehensive research on the electrochemical performance of the cross-linking PAMM as well as PMMA in LiNi0.5Mn1.5O4/Li half-cells at room temperature and 55 °C. Furthermore, the electrochemical properties of LiNi0.5Mn1.5O4/Li/1.5LiTiO2 full cells using this cross-linking PAMM-based electrolyte were presented to demonstrate the huge advantages of high voltage resistance and interfacial compatibility even in harsh conditions provided by the synergetic effect of acrylic anhydride and methyl methacrylate.

### EXPERIMENTAL SECTION

#### Synthesis of PAMM.

All chemicals, battery-grade LiPF6, ethylene carbonate (EC), diethyl carbonate (DEC), methyl methacrylate monomer (MMA), acrylic anhydride, 2-methyl-acrylic acid-2-oxirane-ethyl ester (MAEOE), azobis(isobutyronitrile) (AIBN), and methyl ethyl carbonate (MEC), were purchased from Aldrich and used as received. The PAMM was synthesized in glovebox by an in situ polymerization method. The 60 wt % electrolyte of 1 M LiPF6 in EC/DEC (1:1, v/v) with 30 wt % MMA monomer and 10 wt % acrylic anhydride was prepared and marked as solution A. The electrolyte of MAEOE with 1 wt % AIBN in MEC stored at low temperature was marked as solution B. Solutions A and B were mixed with volume ratio of 1:1 with a water content value of 30 ppm. Free radical reaction and acrylic acid catalyzed ring-opening reaction happened at the onset of blending. LiPF6 in solution A would act with trace of water and generate PF5. The resultant PF5 acted as Lewis acid and then catalyzed ring-opening reaction of MAEOE immediately and formed cross-linking points. In the meantime, the double bond in MAEOE will react with methyl methacrylate and acrylic anhydride with the addition of AIBN at 50 °C. After heating for 6 h, a polymer network of PAMM with multiadvantages was obtained finally. As a contrast, PMMA electrolyte was prepared by mixing MMA monomer, 1 M LiPF6 in EC/DEC (1:1, v/v) and AIBN in ratio of 1:2.5:0.5% poly(ethylene terephthalate) (PET) separators (FPB1511) were used as substrate.

#### Characterization.

Scanning electron microscopy (SEM, Hitachi S-4800 at 2 kV) was taken to observe the surface morphology of the electrolyte and electrodes. Fourier Transform Infrared spectrum (FT-IR, Bruker VERTEX 70) and X-ray diffraction (XRD, Ultima IV) tests were used to evaluate the chemical structures and crystallinity degree of PAMM, respectively. Inductively coupled plasma (ICP, Optima 8300) test was conducted to detect the concentration of dissolved manganese ions in PMMA and cross-linking PAMM-based electrolyte.

X-ray photoelectron spectroscopy (XPS, ESCALab220i-XL) and
transmission electron microscopy (TEM, Tecnai G20, FEI) were used to characterize the surface element composition and surface morphology. Moreover, the thermogravimetric analysis was conducted by METTLER DSC/TGA1 and the water content was tested by Karl Fischer method (870 KF Titrino plus).

The ionic conductivity tests of cross-linking PAMM between two stainless steels was conducted via the electrochemical impedance spectroscopy (EIS) measurement in the frequency range from 1 Hz to 1 MHz. The lithium ion transference number \( t_{Li}^{+} \) was calculated according to following equation

\[
t_{Li}^{+} = \frac{I_0 (\Delta V) - I_S (\Delta V)}{I_0 (\Delta V) - I_S (\Delta V)}
\]

Where \( I_0 \) and \( I_S \) represent initial and steady-state current values, whereas \( R_0 \) and \( R_S \) indicate the resistance of initial and steady state, respectively. \( \Delta V \) is the potential amplitude of 10 mV. The linear sweep voltammetry conducted on a stainless steel working electrode and lithium metal as counter electrode at the scan rate of 1 mV s\(^{-1}\) over the potential range of 0 to 6.0 V versus Li\(^+\)/Li. The cathode pasted on an Al foil was composed of 90 wt % LiNi\(_{0.5}\)Mn\(_{1.5}\)O\(_4\), 4 wt % carbon black, and 6 wt % poly(vinylidene fluoride) (PVDF). The anode was fabricated by blending the active material of Li\(_4\)Ti\(_5\)O\(_12\), carbon black conductive agent and PVDF binder in the weight ratio of 92:3:5. The active material loading was 1.8 mg cm\(^{-2}\). The 2032-type coin cells were assembled where LiNi\(_{0.5}\)Mn\(_{1.5}\)O\(_4\) was used as cathode and Li or Li\(_4\)Ti\(_5\)O\(_12\) as counter electrode, respectively, whereas coin-type LiNi\(_{0.5}\)Mn\(_{1.5}\)O\(_4\)/Li\(_4\)Ti\(_5\)O\(_12\) full cells with PAMM-based gel electrolytes was constructed with N/P values of ca. 1.2 (calculated according to 147 mAh g\(^{-1}\) and 150 mAh g\(^{-1}\) for LiNi\(_{0.5}\)Mn\(_{1.5}\)O\(_4\) and Li\(_4\)Ti\(_5\)O\(_12\), respectively) in an argon-filled glovebox. The gelation reaction occurred at the clicking of mixing solution A and B. The H\(^+\) (PF\(_5\)OH\(^-\))\(^+\) acted as Lewis acid and catalyzed the ring-opening of 2-methyl-acrylic acid-2-oxirane-ethyl ester (MAEOE), and then formed a cross-linking site. Simultaneously, free radical reaction happened between acrylic anhydride, methyl methacrylate, and the double bond of newly generated cross-linking MAEOE and formed a polymer network structure (Figure S1).

The polymerization of acrylic

![Figure 1](image-url)
anhydride, methyl methacrylate and cross-linked MAEOE was
demonstrated by the absence of unsaturated \(-\text{C}==\text{C}−\) groups
and epoxy group in the FTIR spectra (Figure 1b). The
schematic design of PAMM polymer was shown in Figure S1a.
Benefiting from the in situ polymerization process, the obtained
gel polymer electrolyte had smooth and homogeneous surface
morphology and well contact with electrode material (Figure
S1b, c). Furthermore, the cross-linking PAMM-based electro-
lyte had much greater mechanical strength (27.5 MPa) than
that of PMMA polymer (10.2 MPa), as depicted in the stress−
strain curves in Figure 1c.34,35 Moreover, the thermogravimetric
analysis results of PAMM and PMMA were displayed in Figure
S2a, b. The comparison of burning tests on polypropylene
(PP), glass fiber (GF), cellulose separators immersed with
liquid electrolyte and cross-linking PAMM were demonstrated
in Figure S2c, indicating the increase of thermal stability and
the best flame-retarding property of PAMM.

Besides the above merits, the amorphous structure of PAMM
(Figure S3) would also have flexible pathway for lithium
transport. The ion-conductive behavior at different temperature
from 273 to 353 K was depicted in Figure 2a. A distinct
curvature was exhibited especially in the low temperature
region, which can be described by empirical Vogel–Tamman−
Fulcher (VTF) equation:

$$\sigma(T) = A T^{-1/2} \exp\left(\frac{E_a}{R(T - T_0)}\right)$$  \hspace{1cm} (2)

R is a gas constant, A is the pre-exponential factor, \(T_0\) is the
temperature correlated to the glass transition temperature,
which is called equilibrium glass transition temperature, \(E_a\) is
the activation energy. It was worth noticing that the \(E_a\) value
was calculated to be 1 kJ mol\(^{-1}\), a value lower than that of
traditional gel electrolyte (10 kJ mol\(^{-1}\)) and solid electrolyte
(40 kJ mol\(^{-1}\)).29,36,37 A conductivity value of \(6.79 \times 10^{-4}\) S
\(\text{cm}^{-1}\) was achieved at room temperature, which was attributed
to the low energy barrier.38 The high ionic conductivity value

Figure 2. (a) Ionic conductivity dependence with temperature of cross-linking PAMM and (b) Current−time profiles following a DC polarization of 0.01 V of the cross-linking PAMM. Inset is Nyquist profiles of the cell before and after polarization. Comparison of linear sweep voltammograms of PAMM and PMMA at a scan rate of 1 mV s\(^{-1}\) at (c) room temperature and (d) 55 °C, and (e) Chronopotentiometry profiles of PAMM and PMMA in Li/Li symmetric cells.
may result from the soft segments of PMMA and the plasticization of carbonate solvent \(^{39-41}\) and the relevant impedance spectra at different temperature were depicted in Figure S4.

As depicted in Figure 2b, a remarkable lithium ion transference number \( (t_{Li^+}) \) value of 0.51 for cross-linking PAMM was obtained. The high \( t_{Li^+} \) can effectively reduce the polarization in the charging/discharging process, thus favorably improve the power density. In Figure 2c, d, linear sweep voltammetry was used to contrast the electrochemical stability window of PMMA and cross-linking PAMM-based electrolyte at room temperature. PMMA-based one decomposed at around 4.5 V vs Li\(^+/^-\)/Li, whereas cross-linking PAMM had no significant decomposition current up to 5.2 V vs Li\(^+/^-\)/Li. When the temperature was elevated to 55 °C, the PAMM-based electrolyte maintained electrochemical stability up to 5.0 V because of the synergistic effect of the formation of cross-linking structure and the addition of anhydride, which works as anti-high voltage resistant component, while PMMA began to decompose around 4.0 V. It was indicated that the cross-linking PAMM was relatively chemical inert at high voltage of 5 V vs Li\(^+/^-\)/Li, which endowed it very promising polymer electrolyte for high voltage batteries. The linear sweep voltammetry of PAA was also showed in Figure S5 as a comparison, it decomposed at 5.0 V vs Li\(^+/^-\)/Li. The interfacial compatibility of PAMM and PMMA based electrolyte with Li foil was compared by conducting symmetric cells test under a current density of 0.2 mA cm\(^{-2}\) and the result was shown in Figure 2e. The cell with PAMM maintained stable Li plating/stripping during the entire process and demonstrated an extraordinary cycling stability. A small polarization was observed after cycling for 200 cycles and no sign of short cut appeared, indicating a smooth and flat interface of lithium anode, which will be proved in the following sections. By a sharp contrast, PMMA symmetric cell showed a more severely polarization, demonstrating an unstable surface and a poor lithium deposition/stripping. The long cycling performance of PAMM symmetric cell was conducted at density of 0.2 mA cm\(^{-2}\) for 1000 cycles to verify the stable Li plating/stripping in Figure S6. A stable charge and discharge curve was showed and demonstrated a relatively stable interface formed under the effect of “Shuangjian hebi”. In addition, the interfacial chemical compatibility of Li/cross-linking PAMM/Li and Li/PMMA/Li were also evaluated by analyzing the AC impedance variation with elapsed time (Figure S7), which
indicated the high interface stability between cross-linking PAMM and Li metal after 30 days standing.

**Electrochemical Properties of LiNi0.5Mn1.5O4/Cross-Linking PAMM/Li Cells.** To further investigate the feasibility of cross-linking PAMM in high voltage LIBs, the electrochemical behaviors of LiNi0.5Mn1.5O4/Li cells using cross-linking PAMM and PMMA electrolytes cycled between 3.5 and 5.0 V at room temperature were displayed in Figure 3. As shown in Figure 3a, the voltage plateau in both cells was consistent with the typical LiNi0.5Mn1.5O4 charge–discharge profiles. Nevertheless, the voltage gap between charge and discharge profiles of LiNi0.5Mn1.5O4/PMMA/Li cell increased dramatically during cycling, compared with slightly increased LiNi0.5Mn1.5O4/PAMM/Li cell. This result indicated PAMM-based electrolyte can endow a superior interfacial compatibility and suppress the polarization of LiNi0.5Mn1.5O4/Li cell remarkably, which was proved by the improved cycling performance of LiNi0.5Mn1.5O4/PAMM/Li cell (Figure 3b). The first reversible capacity for LiNi0.5Mn1.5O4/PAMM/Li cell was 131.7 mAh g⁻¹. After 500 cycles, the discharge capacity was still 104 mAh g⁻¹, corresponding to the capacity retention of 78.9%. As for LiNi0.5Mn1.5O4/PMMA/Li cell, the first discharge capacity was up to 123.1 mAh g⁻¹ but only 52 mAh g⁻¹ capacity was delivered at the high rate of 5 C. The capacity recovered to 117.6 mAh g⁻¹ in the following cycles at the rate of 0.1 C. The improved rate capability of LiNi0.5Mn1.5O4/PAMM/Li might be ascribed to the flexible lithium ion transfer, the high voltage resistance and the good interfacial compatibility provided by acrylate and anhydride groups in situ polymerized cross-linking PAMM.

Generally, the elevated temperature will facilitate the carbonate solvent decomposition at high potential in LIBs. As a result, the capacity deterioration would be accelerated at elevated temperatures (>50 °C), which was a main obstacle limiting the widespread applications of high-voltage LiNi0.5Mn1.5O4. Therefore, the synergistic effects of PAMM-based electrolyte were further expected at the elevated temperature of 55 °C (Figure S9). As for the cycling characteristics, LiNi0.5Mn1.5O4/PAMM/Li delivered a capacity of 110 mAh g⁻¹, which was much higher than that of 89.5 mAh g⁻¹ for PMMA-based battery. After 30 cycles, the PAMM-based
electrolyte maintained the Coulombic efficiency over 92%, whereas the counterpart only 75% retained (Figure S9). The prior Coulombic efficiency of PAMM-based electrolyte at 55 °C can be attributed to the chemical stability endowed by anhydride and the compatible interface provided by methyl methacrylate. These results proved that PAMM-based electrolyte can significantly enhance the high temperature capacity and cycling stability of LiNi_{0.5}Mn_{1.5}O_{4}/Li cell.

Interface Compatibility of LiNi_{0.5}Mn_{1.5}O_{4}/Cross-Linking PAMM/Li Cell. To further understand the synergistic effects of AA and MMA, we tested the electrochemical impedance spectroscopy (EIS) results (Figure 4) of fully discharged cells of LiNi_{0.5}Mn_{1.5}O_{4}/cross-linking PAMM/Li and LiNi_{0.5}Mn_{1.5}O_{4}/ PMMA/Li. The corresponding fitting results and equivalent electric circuit simulated with Z Simp Win software were recorded in Table S1. It showed that the cell impedance with PMMA based electrolyte amplified much more (∆R_{SEI} ≈ 51 Ω and ∆R_{CT} ≈ 160 Ω) than that of PAMM-based cell (∆R_{SEI} ≈ 38 Ω and ∆R_{CT} ≈ 98 Ω) after 500 cycles. The high interface impedance of LiNi_{0.5}Mn_{1.5}O_{4}/PMMA/Li may leading to the serious polarization during cycling and caused the poor battery performance in Figure 3. This result also indicated the interface between electrodes and PAMM-based electrolyte was much more compatible than that in LiNi_{0.5}Mn_{1.5}O_{4}/PMMA/Li cell, which might be owing to the less erosion of electrolyte and the alleviated Mn dissolution by virtue of the synergistic effect of AA and MMA.

The typical SEM images of pristine Li foil, Li foil in PAMM half-cell and PMMA half-cell after 500 cycles were showed in Figure 5a−c, respectively. The Li foil in PAMM half-cell exhibited a smooth and compact surface with little cracks after 500 cycles, whereas the Li foil in PMMA half-cell showed severe impurity accumulation and incompact surface. For high voltage spinel cathode material, the pristine and the cycled LiNi_{0.5}Mn_{1.5}O_{4} electrodes with PAMM and PMMA based electrolytes demonstrated typical spinel crystal morphology (Figure 5d, g). The mapping result (Figure 5e, f, Table S2) of LiNi_{0.5}Mn_{1.5}O_{4} electrode faced with PAMM showed the same O and Mn element distribution. In comparison, for LiNi_{0.5}Mn_{1.5}O_{4} electrode assembled with PMMA electrolyte, the Mn element distributed in the whole observation area rather than in the specific area just like O element did, suggesting the serious Mn dissolution from LiNi_{0.5}Mn_{1.5}O_{4}. The ICP-MS analysis (Table S3) further confirmed the dissolution of Mn element. Mn ion concentration of 56.2 and 11.0 ppm was obtained at 25 °C for PMMA and cross-linking PAMM-based electrolyte, respectively, which reflected a more severe Mn dissolution in PMMA-based system after cycling (Figure 5i). The corresponding value of Mn ion concentration at 55 °C was also listed in Table S3, which demonstrated that the higher the temperature, the more Mn ions dissolved in the PMMA-based cell. The above results indicated that the cross-linking PAMM could not only retard the dissolution of transition metal at severe conditions such as high potential and high temperature but also enhance the interface stability between electrolyte and Li metal, which confirmed the synergetic effect of AA and MMA.

To further understand the origin of the above difference in impedance evolution and element distribution between PAMM- and PMMA-based electrolyte involved LNMO electrodes, interface composition of the two electrodes after 500 cycles were studied by XPS in Figure 6. At pristine state, O 1s and C 1s spectra show characteristic peaks of LiNi_{0.5}Mn_{1.5}O_{4} electrode. In F 1s spectra for two electrodes (Figure 6a), the peak values of 685.2, 686, 687, and 688.1 eV belong to LiF, Li_{x}PF_{y}O_{z}, CP_{2}, and C−F, respectively. The intensity of LiF increased greatly for PMMA-based electrode, indicating the much severe electrolyte decomposition. In O 1s and C 1s spectra, the lattice oxygen peak decreased while the oxygen−carbon peaks increased, demonstrating the formation of cathode electrolyte interphase (CEI) layers on the surface of both electrodes after
The peak intensity at 530.4–531.94 eV of both electrodes is regarded as the formation of semicarbonate (ROCO₂Li) and carbonate (Li₂CO₃) species as well as the ester group with carbonate oxygen from decomposition of carbonate solvent. In the O 1s branch, the spectrum of the PMMA exhibits a more clearly grown peak at around 530 eV, indicating more significant decomposition of carbonate solvent. The formation of CEI could equally be deduced from C 1s. The intensity of carbonyl group and alkyl group with carbonate oxygen increased because of the decomposition of carbonate solvent. These results shed light on the CEI composition being similar for two electrodes, and the moderate thickness and integrity of the CEI layer should have a significant effect on the electrode performance.

The TEM images (Figure 7a, b) of the cycled LiNi₀.₅Mn₁.₅O₄ electrode with PAMM-based electrolyte displayed an amorphous surface layer known as CEI with a thickness approximately 10 nm, which is more uniform and compact than that of the LiNi₀.₅Mn₁.₅O₄ electrode with PMMA electrolyte (Figure 7c and 7d). Furthermore, the covering film on PMMA involved LiNi₀.₅Mn₁.₅O₄ showed a cracked and heterogeneous morphology, indicating an unstable CEI formation. In a conclusion, the intact and robust CEI formed on LiNi₀.₅Mn₁.₅O₄ electrode in PAMM-based electrolyte could facilitate the transfer of lithium ions, alleviate the erosion of electrolyte, repress the transition metal dissolution in cathode, and then significantly improve the performance of LIBs.

The mechanism of “Shuangjian hebi” electrolyte system alleviated Mn dissolution may be interpreted as follows. The fully charged state LiNi₀.₅Mn₁.₅O₄ would catalyze the decomposition of solvent in PMMA based electrolyte, in return, the dissolution of transition metal and the disproportioned reaction of Mn²⁺ would be accelerated by electrolyte oxidation. Hence, when cross-linking PAMM-based polymer network was introduced to replace the electrolyte-swelling PMMA-based electrolyte, the electrolyte decomposition will be
Li$_4$Ti$_5$O$_12$. As presented in Figure 9a, the LiNi$_{0.5}$Mn$_{1.5}$O$_4$/PAMM/Li$_4$Ti$_5$O$_12$ half cell were superior than that of the LiNi$_{0.5}$Mn$_{1.5}$O$_4$/Li$_4$Ti$_5$O$_12$ full cell. Because the PAMM-based electrolyte had prominent interface stability with LiNi$_{0.5}$Mn$_{1.5}$O$_4$ and Li metal as mentioned above, in order to investigate the fading reason for full cell, EIS spectra of LiNi$_{0.5}$Mn$_{1.5}$O$_4$/PAMM/Li half-cell, LiNi$_{0.5}$Mn$_{1.5}$O$_4$/PAMM/Li$_4$Ti$_5$O$_12$ full cell and Li$_4$Ti$_5$O$_12$/PAMM/Li half-cell were shown in Figure 9b. Interfacial resistance for each cell increased with cycling and the Li$_4$Ti$_5$O$_12$ half-cell displayed a much faster increase compared to that of the LiNi$_{0.5}$Mn$_{1.5}$O$_4$ half cell. The interaction between Li$_4$Ti$_5$O$_12$ and PAMM will be discussed in our further work to dilute the reason for faster increase resistance. It can be deduced that the interfacial impedance of anode side made a greater contribution to the impedance increase as well as the capacity fading of the full cell. It also elucidated the reason why the properties of LiNi$_{0.5}$Mn$_{1.5}$O$_4$ half-cell were superior than that of the LiNi$_{0.5}$Mn$_{1.5}$O$_4$/Li$_4$Ti$_5$O$_12$ full cell.

**CONCLUSIONS**

In summary, we proposed a strategy to solve the challenges of LiNi$_{0.5}$Mn$_{1.5}$O$_4$ high voltage battery system by in situ generated cross-linking poly(acrylic anhydride-2-methyl-acrylic acid-2-oxirane-ethyl ester-methacrylate) (PAMM) electrolyte inspired by “shuangjian hebi” concept. The cross-linking PAMM-based electrolyte not only integrated the advantages of both acrylic anhydride and methyl methacrylate, including high ion conductivity at room temperature (6.79 × 10$^{-4}$ S cm$^{-1}$), broad voltage window (above 5 V), stable interfacial compatibility with Li metal, but also overcome the shortcoming of low mechanical strength of PMMA. Furthermore, the in situ polymerization of two monomer solutions in battery formed excellent contact and superior binding between PAMM-based electrolyte and both electrodes. As a result, the LiNi$_{0.5}$Mn$_{1.5}$O$_4$/Li cell with PAMM-based electrolyte showed notably improved cycling stability, Coulombic efficiency, and rate capability at room temperature and 55 °C when compared with the battery used PMMA-based electrolyte. The EIS, ICP, and SEM results rationally demonstrated that the PAMM formed a more stable passivation layer on the cathode surface, which was ion conductive and effectively resistant to electrolyte decomposition and transition metal dissolution owing to the synergistic effects of anhydride and acrylate groups. The performance of LiNi$_{0.5}$Mn$_{1.5}$O$_4$/PAMM/Li$_4$Ti$_5$O$_12$ full cell further proved the feasibility of PAMM-based electrolyte in high voltage battery system. A capacity of 128 mAh g$^{-1}$ and a significantly improved Coulombic efficiency of 96% at 0.1 C after 100 cycles for LiNi$_{0.5}$Mn$_{1.5}$O$_4$/Li$_4$Ti$_5$O$_12$ were first presented. Although further insight work is still in progress for elucidating the interfacial evolution, these findings demonstrated that the high ion conductivity and wide voltage window of PAMM-based electrolyte. The capacity of 128 mAh g$^{-1}$ was delivered in the first cycle and 113.2 mAh g$^{-1}$ remained with a Coulombic efficiency of 98.9% after 100 cycles, which was better than that of the full cell with commercial liquid electrolyte (1 M LiPF$_6$ in EC-DEC) (Figure S10) as well as the previously reported battery system. In contrast, the cycling property of full cell with PAMM-based electrolyte at 55 °C was measured and showed in Figure S11, which delivered a capacity of 91.8 mAh g$^{-1}$ after 23 cycles. The severe capacity fading may be ascribed to the aggravated catalytic decomposition of EC and DEC by Ti$^{4+}$ at high temperature. From the rate performance measurement, the first cycle reversible capacity of 129 mAh g$^{-1}$ was delivered at 0.1 C and 50 mAh g$^{-1}$ was remained at 5 C (Figure S12). Because the PAMM-based electrolyte had prominent interface stability with LiNi$_{0.5}$Mn$_{1.5}$O$_4$ and Li metal as mentioned above, in order to investigate the fading reason for full cell, EIS spectra of LiNi$_{0.5}$Mn$_{1.5}$O$_4$/PAMM/Li half-cell, LiNi$_{0.5}$Mn$_{1.5}$O$_4$/PAMM/Li$_4$Ti$_5$O$_12$ full cell and Li$_4$Ti$_5$O$_12$/PAMM/Li half-cell were shown in Figure 9b. Interfacial resistance for each cell increased with cycling and the Li$_4$Ti$_5$O$_12$ half-cell displayed a much faster increase compared to that of the LiNi$_{0.5}$Mn$_{1.5}$O$_4$ half cell. The interaction between Li$_4$Ti$_5$O$_12$ and PAMM will be discussed in our further work to dilute the reason for faster increase resistance. It can be deduced that the interfacial impedance of anode side made a greater contribution to the impedance increase as well as the capacity fading of the full cell. It also elucidated the reason why the properties of LiNi$_{0.5}$Mn$_{1.5}$O$_4$ half-cell were superior than that of the LiNi$_{0.5}$Mn$_{1.5}$O$_4$/Li$_4$Ti$_5$O$_12$ full cell.
PAMM-based electrolyte was a very promising candidate toward the batteries with high working potential and even at high operational temperature.

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